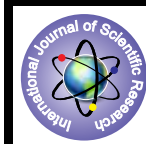


Dielectric Behavior of Perovskite Barium Titanate and Barium Strontium Titanate Ceramics



Chemistry

KEYWORDS : BST ceramics, sol-gel technique, dielectric behaviour

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ABSTRACT

Stoichiometric compositions [(Ba+Sr):Ti = 1:1] of barium strontium titanate (Ba_{1-x}Sr_xTiO₃) (where molar concentration $x = 0, 0.2, 0.3, 0.4, 0.5$ and 1.0) powders were prepared using a sol-gel technique via acetate method. The as-fired amorphous powders were annealed at 850°C to form nanocrystalline powders. These powders were then cold-pressed to make pellets which were then sintered at 1300°C to form dense ceramics. These ceramic pellets were electroded using conductive silver paste and characterized for their frequency and temperature dependent dielectric behavior in order to investigate the effect of strontium concentration on the dielectric constant of barium titanate (BaTiO₃). It was observed that with increasing strontium concentration the ferroelectric to paraelectric phase-transition temperature (T_c) shifted to lower temperatures and dielectric constant increased substantially. With a very small addition of Sr (Ba_{0.8}Sr_{0.2}TiO₃), a high dielectric constant and a low dielectric loss (<0.1) were obtained at room temperature (25°C). The dielectric constant of BST ceramics measured at room temperature as a function of frequency did not change much with frequency and their magnitude was almost same at higher frequencies (1 kHz to 100 kHz).

1. Introduction

The choice of capacitors for memory applications depends on high thermal stability, high fatigue resistance, high dielectric constant, low dielectric loss, high charge storage capacity, and low leakage current density of the dielectric material. Barium titanate (BaTiO₃) possesses all these characteristics and hence it is regarded as one of the promising material to replace SiO₂ as dielectric material for such applications [1,2]. Demand for lead free dielectric and ferroelectric materials have further increased the significance of barium titanate based thin films and ceramics. Ferroelectric oxides crystallizing in perovskite structure possess relatively simple crystal structure that has the ABO₃ stoichiometry. Typically A site cation is large and is coordinated by 12 anions in the lattice. The B-site cation is smaller- frequently it is a transition metal being six coordinated forming BO₆ octahedral. The distortions of this octahedral leads to deviations from perfect cubic symmetry [3,4]. Barium Strontium Titanate (BST) Ba_{1-x}Sr_xTiO₃ which is known to adopt the ABO₃-type structure is a continuous solid solution of BaTiO₃ and SrTiO₃ over the whole concentration range. In this case Ba and Sr are coordinated by 12 oxygen atoms and Sr is octahedrally coordinated with 6 oxygen atoms [5,6]. The properties of Ba_{1-x}Sr_xTiO₃ depend on the composition x . Partial substitution of Ba ions in pure BaTiO₃ strongly affects the ferroelectric-paraelectric phase transition temperature. SrTiO₃ is usually added as a shifter in order to move the Curie point T_c to lower temperatures. It is well established that T_c of barium titanate decreases linearly with the amount of Sr²⁺ in place of Ba²⁺. For bulk Ba_{1-x}Sr_xTiO₃ ceramics, the Curie point varies from 120°C to -240 °C, whereas the relative electric permittivity (measured at room temperature and electric field of frequency $\nu = 1$ kHz) decreases from $\epsilon' \sim 2000$ to $\epsilon' \sim 300$ for x from $x = 0.0$ to $x = 1.0$. As a result, the transition temperature, and hence the electrical and optical properties of Ba_{1-x}Sr_xTiO₃, can be tailored over a broad range to meet the requirements of various electronic applications. Among the various methods used to prepare the BST powders, sol-gel processes are commonly applied in the preparation of BST fine powders and thin films due to the precise control of composition, ease of homogenous distribution of elements, cost effectiveness, relatively low temperatures and simple process requirements [7]. For a single phase cubic perovskite structure of Ba_{0.5}Sr_{0.5}TiO₃

achieved at a calcination temperature of 800°C an optimum dielectric constant of 1164, dielectric loss of 0.063 and the lowest leakage current density of 49.4 pA/cm² at 5V at room temperature have been reported [8]. In this work, the frequency and temperature dependent dielectric behavior of BST ceramics (Ba_{1-x}Sr_xTiO₃ where molar concentration $x = 0, 0.2, 0.3, 0.4, 0.5$ and 1.0) prepared using sol-gel derived powders is being reported. These ceramics were characterized for their structural, morphological, dielectric and electrical properties. The main objective was to study the effect of A-site Sr substitution on the dielectric properties of BT.

2.0 Materials and Methods

Table 1 shows the properties and source of the various chemicals used in the present synthesis. The starting materials were Barium acetate [Ba(CH₃COO)₂ ≥ 99%], Strontium acetate [Sr(CH₃COO)₂] and Titanium tetrabutoxide (97%). Glacial acetic acid [CH₃COOH ≥ 99.7%] and butyl alcohol [C₄H₉OH > 99.5%] were used as solvents. Calculated amounts of Barium acetate and strontium acetate were weighed to synthesize stoichiometric compositions of (Ba_{1-x}Sr_xTiO₃) (where molar concentration $x = 0, 0.2, 0.3, 0.4, 0.5$ and 1.0) [(Ba+Sr):Ti = 1:1] powders. The weighed amount of these compounds were mixed together in glacial acetic acid and refluxed at 100 °C for 0.5h using a sand bath. The Ba-Sr solution was cooled down to the room temperature and then the calculated amount of Titanium Tetrabutoxide mixed with butyl alcohol was added and the resulting solution was stirred again using a magnetic blender for 0.5h at room temperature. After stirring, the sol was hydrolysed using 10-15 ml of distilled water. Small amount of acetylacetone [CH₃C(O)CH₂C(O)CH₃] was added as stabilizer. The sol was relatively clear and stable. It was covered with aluminium foil and kept in a dessicator to avoid air contamination. It was left in open for gelation and it gelled in 3 days. The obtained gel was fired at 350 °C in air for 3-4 hrs. The as-fired powder obtained was amorphous in nature. This powder was calcined at 850°C for 3h in a muffle furnace to obtain a poly-crystalline powder. The BST powder so obtained was milled and die-pressed into thick pellets under a pressure of 300 MPa. The samples were finally sintered at 1300°C for 3h at a heating rate of 100°C/min. The powder and ceramics were prepared using a procedure given in the

flowchart (Fig. 1). The sample designation and compositions are listed in Table 2.

2.1 Material Characterization Structural characterization of barium titanate and barium strontium titanate (BST 73) powder calcined at 850°C was done using X-Ray Diffractometer. Diffractograms of BT powder, BST powder and BST ceramic are shown in Fig. 2. The peaks show a polycrystalline nature and confirm the formation of tetragonal structure of BT and BST.

2.1.1 Dielectric studies

AC frequency dependent values of parallel capacitance (C_p) and loss tangent or dissipation factor ($\tan\delta$ or D) of the ceramic pellets were measured by using Agilent 4263B precision LCR meter. The actual thickness of samples was measured using a micrometer before electroding. Electroding was done using conductive silver paste using a paint brush. Electroded samples were placed in a sample cell between two electrodes and voltage was applied. The parallel capacitance (C_p) was measured on the LCR meter. Dielectric constant (ϵ') was calculated using the formula:

$$\epsilon' = C/C_0 \quad \dots\dots\dots(1)$$

where, C_0 ($=\epsilon_0 A/d$) is the capacitance with vacuum between parallel plates, and ϵ_0 ($=8.85 \times 10^{-12}$ F/m) is the permittivity of free space, A is the area of electrodes, and d is the thickness of the sample. Dissipation factor (D or $\tan\delta$) was directly measured on the LCR meter. To calculate ac conductivity, parallel capacitance (C_p) and dissipation factor (D or $\tan\delta$) from 100 Hz up to 100 kHz was measured. $\sigma_{ac}(\omega)$ was calculated by using the following expression:

$$\sigma_{ac}(\omega) = 2\pi f \epsilon_0 \epsilon'' = 2\pi f \epsilon_0 \epsilon' \tan\delta \quad \dots\dots\dots(2)$$

where, $\tan\delta$ is the dissipation factor, $\omega=2\pi f$ is the angular frequency, ϵ_0 is the permittivity of free space ($=8.865 \times 10^{-12}$ F/m), ϵ' is the dielectric constant ($\epsilon''/\epsilon' = \tan\delta$), and ϵ'' is the dielectric loss.

Temperature dependent and frequency dependent dielectric parameters were measured from room temperature (30°C) to 200°C at 100 Hz, 120 Hz, 1 kHz, 10 kHz and 100 kHz.

3.0 Results and discussion

Temperature dependent dielectric constant and dielectric loss of all the ceramics measured at 1 kHz are shown in Fig 3a and Fig 3b respectively. From Fig. 3a it is clear that the T_c of BaTiO₃ (~120°C) has shifted to the lower temperatures (<30°C) for BST samples with addition of Sr content. Furthermore, values of dielectric constant decrease sharply with increase in temperature in paraelectric phase. These values are in accordance with the reported literature according to which the dielectric constant can be varied by changing the Ba/Sr ratio [9]. It has been reported that dielectric constant increases with decrease in Sr content. Highest dielectric constant is obtained at room temperature when (Ba+Sr)/Ti ratio is 1:1 [10]. BST composition (Ba_{0.8}Sr_{0.2}TiO₃) exhibits the highest dielectric constant and a T_c ~30°C which is closer to the room temperature. SrTiO₃ shows lowest dielectric constants at all temperatures. Figure 3b shows that the dielectric loss is very small (<1) for all the samples minimum being that for BaTiO₃ and Ba_{0.5}Sr_{0.5}TiO₃. Figure 4a and 4b show the frequency dispersion of dielectric constant (ϵ') and dielectric loss (ϵ'') respectively of all the samples measured at room temperature (25°C). From Fig. 4a it can be seen that at higher frequencies (>1 kHz) a very slight change is observed in the values of dielectric constant. (Ba_{0.8}Sr_{0.2}TiO₃) shows highest values of dielectric constant at all frequencies. Similarly, there is no or little variation of dielectric loss at higher frequencies (>1 kHz) for all the samples except SrTiO₃ and Ba_{0.6}Sr_{0.4}TiO₃. Figure 5 shows the temperature dependent

dielectric constant of barium titanate ceramics at different frequencies. It is observed that T_c for BaTiO₃ is almost constant at 120°C at all frequencies. Below T_c (ferroelectric phase), dielectric constant is distinguishable at different frequencies. However near and above T_c (paraelectric phase) the magnitude of dielectric constant remains same and independent of frequency. Figure 6 shows the variation of dielectric constant as a function of mole fraction of Sr. With small addition of Sr content ($x = 0.2$) a very high value of dielectric constant and a lower T_c are obtained at room temperature.

4.0 Conclusions

Bulk ceramics of Barium strontium titanate (Ba_{1-x}Sr_xTiO₃) prepared using fine powders derived from sol-gel method showed that with increasing strontium concentration the ferroelectric-paraelectric phase transition temperature (T_c) shifts to lower temperatures. With a very small addition of Sr (Ba_{0.8}Sr_{0.2}TiO₃), a very high dielectric constant ($\epsilon_{max}>2000$) and a low dielectric loss (<0.1) is obtained at room temperature (25°C). The dielectric constant of BST ceramics shows no frequency dispersion at higher frequencies (1 kHz to 100 kHz).

Acknowledgements

Authors wish to express gratitude to University Grants Commission (UGC) for the funding of this project. (NSR) and (MC) are grateful to UGC for a project fellowship.

S.No	CHEMICAL NAME	MOLECULAR FORMULA	MOLECULAR WEIGHT(g/mol)	CAS No.	PURITY/ ASSAY (%)
1.	Barium acetate	Ba(CH ₃ COO) ₂	255.42	543-80-6	≥99
2.	Strontium acetate	Sr(CH ₃ COO) ₂	205.71	543-94-2	-
3.	Acetic acid	CH ₃ COOH	60.05	64-19-7	≥99.7
4.	Titanium tetrabutoxide	Ti(OC ₄ H ₉) ₄	340.32	5593-70-4	97
5.	Butanol	C ₄ H ₉ OH	-	71-36-3	>99.5
6.	Acetylacetone	CH ₃ COCH ₂ COCH ₃	100.12	123-54-6	≥99

Table 2 :Sample designation and composition(Ba_{1-x}Sr_xTiO₃)

Sample designation	Sample name	Ba(1-x)	Sr(x)
BT	BaTiO ₃	1.0	0.0
BST82	Ba _{0.8} Sr _{0.2} TiO ₃	0.8	0.2
BST73	Ba _{0.7} Sr _{0.3} TiO ₃	0.7	0.3
BST64	Ba _{0.6} Sr _{0.4} TiO ₃	0.6	0.4
BST55	Ba _{0.5} Sr _{0.5} TiO ₃	0.5	0.5
ST	SrTiO ₃	0.0	1.0

FIG. 1 - FLOWCHART FOR THE PREPARATION OF BARIUM

STRONTIUM TITANATE USING SOL-GEL METHOD



Fig. 2 XRD diffractograms of Barium titanate and Barium Strontium titanate

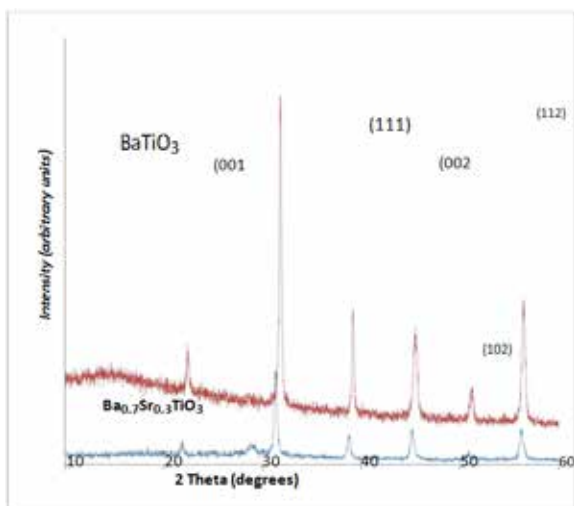


Fig 3a: Influence of Sr⁺⁺ substitution on temperature dependent dielectric constant (measured at 1 kHz) of Barium titanate ceramics sintered at 1300oC

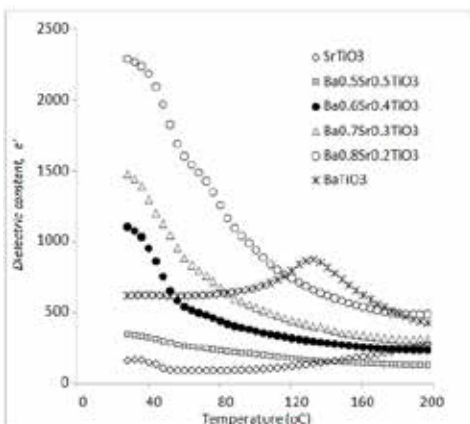


Fig 4a: Influence of Sr⁺⁺ substitution on room temperature

(30oC) dielectric constant (measured at different frequencies) of Barium titanate ceramics sintered at 1300oC

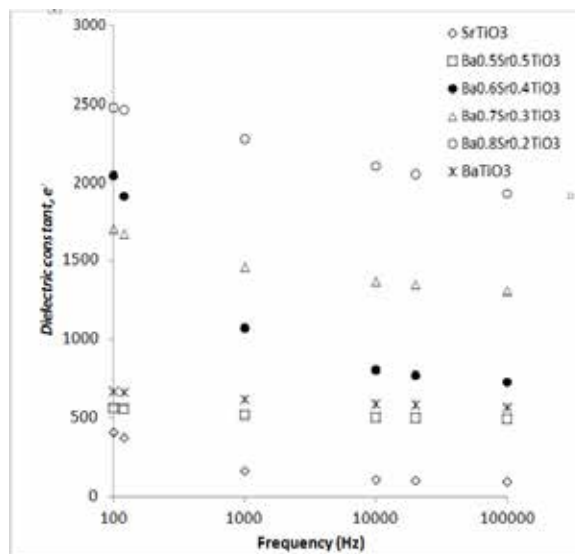


Fig 4b: Influence of Sr⁺⁺ substitution on room temperature (30oC) dielectric loss (measured at different frequencies) of Barium titanate ceramics sintered at 1300oC

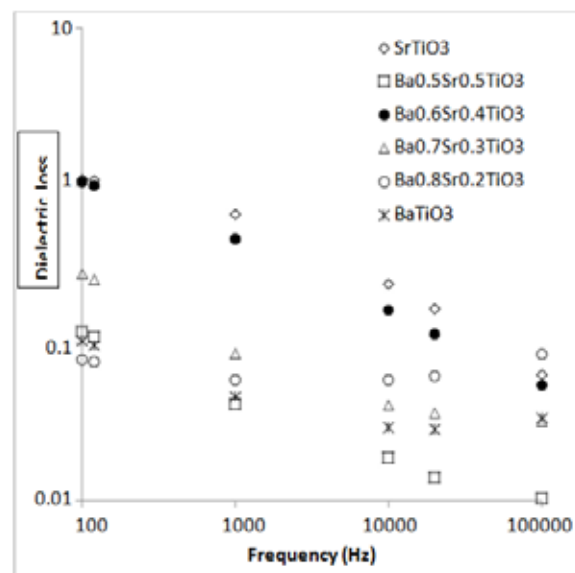


Fig 3b: Influence of Sr⁺⁺ substitution on temperature dependent dielectric loss (measured at 1 kHz) of Barium titanate ceramics sintered at 1300oC

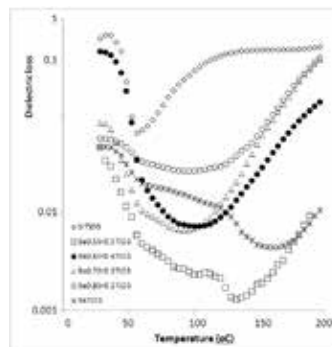


Fig 5: Temperature and frequency dependent dielectric con-

stant of Barium titanate ceramic sintered at 1300oC

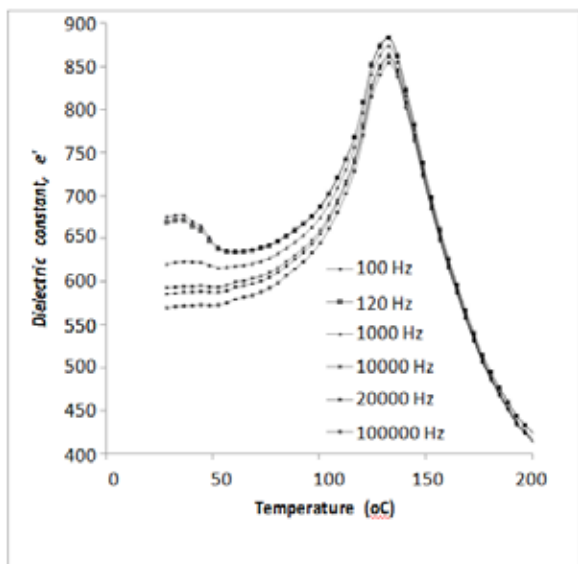
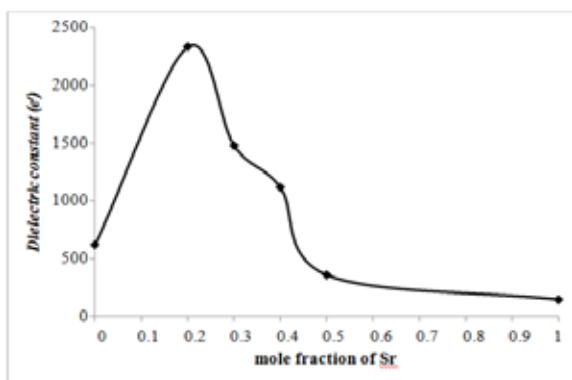


Fig. 6 : Variation of dielectric constant (measured at 1 kHz at room temperature 25oC) of barium titanate as a function of the mole fraction of strontium



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